

1 **PHOTOELECTROCATALYTIC DEGRADATION OF**
2 **METHYLENE BLUE USING ZNO NANORODS FABRICATED ON**
3 **SILICON SUBSTRATES**

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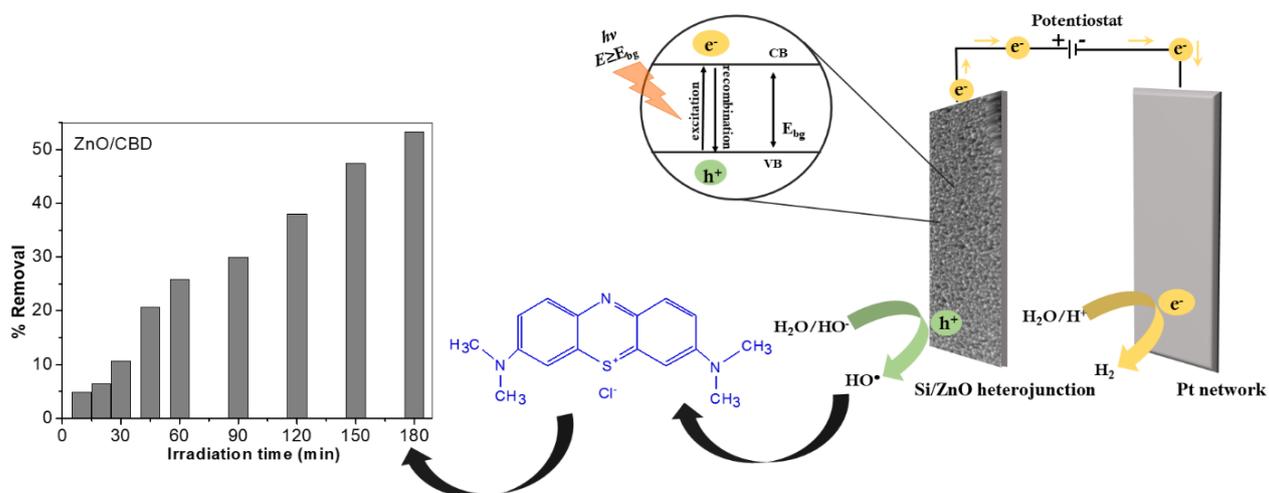
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26 **Graphic Abstract**

27

28 **Abstract**

29 ZnO nanorods were grown on silicon (Si) substrates by two techniques: (i) Chemical Bath
 30 Deposition (CBD) and (ii) a CBD seed layer combined with Carbothermal Reduction
 31 Vapor Phase Transport (CTR-VPT). The structured ZnO nanorods were characterized by
 32 scanning electron microscopy (SEM), X-ray diffraction (XRD), X-ray photoelectron
 33 spectroscopy (XPS), and contact angle measurements. The photoelectrochemical property
 34 of ZnO nanorods were analyzed by linear voltammetry under UV-ABC light excitation.
 35 Using the ZnO nanorod samples as photoanodes, the removal of methylene blue (MB) as
 36 a representative organic compound was studied by the photoelectrocatalytic (PEC)
 37 technique applying a potential (E) of 0.6 V. For comparison purposes, experiments were
 38 performed under the same conditions using photocatalysis (PC), direct photolysis and
 39 using samples of pure Si (support material) as working electrodes in PEC. XRD analyses
 40 of ZnO prepared by both methods showed the expected ZnO wurtzite phase and a
 41 preferred c-axis orientation in the growth of the nanorods. The presence of ZnO was
 42 further confirmed by XPS and contact angle measurements showed that ZnO grown by
 43 CBD (ZnO/CBD) had a slightly hydrophobic behavior while ZnO grown by CTR-VPT
 44 (ZnO/CTR-VPT) is hydrophilic. Both ZnO sample types were shown to be photoactive,

45 with ZnO/CBD showing higher resultant photocurrent compared to ZnO/CTR-VPT. For
46 the degradation of MB 53% of the compound was removed using ZnO/CBD as a working
47 electrode, while using the ZnO/CTR-VPT electrode led to a removal of 43% of MB.
48 However, direct photolysis alone removed 39% of the MB. The lower removal of MB
49 using ZnO/CTR-VPT samples was related to surface dissociation during the degradation
50 process. The results show that ZnO nanorods prepared by the CBD technique are a
51 promising photoelectrode for PEC applications. Our data also indicate that CTR-VPT-
52 grown nanorods produce uniform nanorod arrays, but this uniform nanostructure deposit
53 does not lead to any increase in PEC activity.

54

55 **Keywords:** chemical bath deposition, ZnO nanorods, Si/ZnO heterojunction,
56 photoelectrocatalysis, methylene blue.

57

58 1. Introduction

59 The development of new materials that can be used as semiconductors in
60 photocatalysis (PC) has been the focus of many studies¹. Trends in photocatalysis
61 research are currently focused on the development of ordered semiconductor
62 nanostructures² such as needle-like, nanotubes, nanowires, nanofibers, nanorods, and
63 nanowalls which show higher efficiency due to various factors, including their relatively
64 high surface area and excellent electronic transport properties.¹

65 There are many reports in the literature on the use of various semiconductors in
66 PC. Among them, the most extensively studied is titanium dioxide (TiO₂). On the other
67 hand, zinc oxide (ZnO), with a direct bandgap of ~3.3 eV,³ is an important multi-
68 functional semiconductor, which has clear material advantages such as good
69 photocatalytic activity, low toxicity, high electron mobility, high chemical and thermal

70 stability, a large optical absorption coefficient, and relatively facile synthesis in various
71 nano-structural forms suitable for a diverse range of applications.⁴⁻⁷ Therefore ZnO is a
72 promising potential alternative semiconductor material to replace TiO₂, especially for use
73 in photocatalysis.⁸⁻¹⁰

74 The preparation of different self-organized nanostructures of ZnO on a variety of
75 substrates has attracted much attention to improve the photocatalytic efficiency.¹¹ The
76 growth of ZnO nanostructures is very sensitive to the synthesis parameters, including
77 temperature, pressure, substrate, and gas flow. One of the main challenges in ZnO
78 nanostructure deposition is control of the synthesis to achieve ordered and uniform
79 growth of a particular desired nano-morphology reproducibly over a suitable substrate
80 area.¹¹ It is worth mentioning that the larger the area of the material, the more complex is
81 the uniform and reproducible deposition over the entire surface.

82 ZnO can be grown in various morphology nanostructures, including nanorods,¹²
83 nanobelts,¹³ nanosheets,¹⁴ nanotubes,¹⁵ nanoflowers,¹⁶ nanodisks,¹⁷ amongst others.
84 Aligned nanorod arrays, deposited uniformly over substrate areas, are of great interest in
85 the present study for use as a photoanode, due to their large surface to volume ratio,
86 relatively simple synthesis and the enhancement of light absorption due to multiple light
87 scattering among the ZnO nanorods.¹⁸

88 These aligned nanorod arrays have been produced using a number of growth
89 methods including chemical bath deposition (CBD),¹⁹ vapour phase transport (VPT),²⁰
90 chemical vapour deposition (CVD),²¹ carbothermal reduction vapour phase transport
91 (CTR-VPT),²² electrodeposition,²³ and hydrothermal deposition.²⁴

92 Among these methods, CBD has achieved prominence because of its advantages
93 such as simplicity, controllability, potential for scalable deposition, low cost, and low
94 temperature processing which provides the possibility of using cheap substrates such as

95 plastics, as well as glass or silicon.^{12,25,26} The formation of ZnO nanorods by CBD is
96 explained in detail by Byrne et al.¹⁹

97 The use of CBD seed layers in combination with high temperature nanorod growth
98 using carbothermal reduction VPT (CTR-VPT) is an effective route for preparing higher
99 optical quality nanorods.²² In the present study we grew vertical ZnO nanorod arrays on
100 silicon (Si) substrates using both CBD and CTR-VPT and studied their potential for use
101 as semiconductors in PC and photoelectrocatalysis (PEC).

102 PEC is extremely attractive for applications concerning oxidization of organic
103 compounds.²⁷ In comparison to PC, it is highly efficient and sustainable and does not
104 cause secondary pollution.⁴ It is based on a semiconductor photoanode that is irradiated
105 by light with energy equal or greater than its band gap and simultaneously with the
106 application of an external bias potential on the semiconductor. When the potential is
107 applied a reduction of the recombination the photo-generated electron-hole pairs occurs
108 and consequently an increase of the PC efficiency is obtained.^{28,29}

109 Nevertheless, at the best of our knowledge, only a few studies are found in the
110 literature using ZnO nanostructures as a photoanode for PEC, including the studies of
111 Hunge et al.⁶, Han et al.³⁰, Liu et al.³¹, Lin et al.³², Suryavanshi et al.³³, and Sarwar et al.³⁴
112 These employ nanostructured ZnO as photoanodes for PEC. In addition, only the work of
113 Han et al.³⁰ used ZnO synthesized by CBD as a photoanode for PEC applications. It is
114 also worth mentioning that, for applications in PEC, no work has been reported in the
115 literature using ZnO grown on (inexpensive and widely available) Si substrates. Hoa et
116 al.³⁵ fabricated ZnO nanorods on glass and Si substrates by a hydrothermal method and
117 investigated the PC activity of this structure for degradation of Rhodamine B under UV
118 light irradiation. The results indicated that the Si/ZnO nanorod heterojunction exhibits
119 higher photocatalytic activity compared to that of a glass/ZnO nanorod junction. Among

120 various hybrid forms, when a ZnO/Si heterojunction is formed, photogenerated charge
121 carriers can be separated by an internal electric field and consequently photocarrier
122 recombination is inhibited, contributing to the improvement of PC activity.^{35,36} Therefore,
123 further investigations of the Si/ZnO nanorod heterojunction in PEC is a promising area
124 of research, which remains largely unexplored.

125 The aim of this work is to prepare ZnO nanorods on Si by both the CBD and CTR-
126 VPT methods and to utilise the prepared ZnO deposits for PEC degradation, using
127 methylene blue (MB) as a model organic compound, comparing the efficiency of both the
128 materials synthesised by the two techniques. The crystallinity, surface morphology and
129 PEC performance of the deposited ZnO nanorods were investigated with various
130 techniques such as scanning electron microscopy (SEM), X-ray diffraction (XRD), X-ray
131 photoelectron spectroscopy (XPS), contact angle and linear voltammetry measurements.

132

133 **2. Material and Methods**

134

135 *2.1. Materials*

136 P-type silicon wafers with a (100) surface orientation were purchased from
137 Wacker-Chemitronic GMBH. Zinc acetate ($\geq 99\%$) was purchased from Riedel-de Haën.
138 ZnO and graphite, both of high purity ($\geq 99.9\%$) were provided by Alfa Aesar. MB (97%
139 pure) was purchased from Dinâmica do Brasil. The other reagents and solvents
140 (purchased elsewhere) were used as received. Deionized water was used in all
141 experiments.

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145 *2.2.Synthesis of ZnO nanorod arrays*

146 ZnO nanorods were grown on silicon (Si) substrates by two techniques: (i) CBD
147 and (ii) CTR-VPT. Both procedures were based on the reports of Byrne et al.^{19,22}

148

149 2.2.1. Chemical Bath Deposition (CBD)

150

151 Silicon (100) wafers were cleaved into small rectangles with an area of 6.5 cm²,
152 and were then cleaned by sonication in ethanol and dried in a nitrogen stream.

153 First, a thin ZnO seed layer was formed on the substrate. For this, 24.375 μL of a
154 5 mmol L⁻¹ zinc acetate solution prepared in absolute ethanol was deposited by drop
155 coating on the center of the substrate. This amount is equivalent to 3.75 μL of zinc acetate
156 per cm² of substrate, according to the previous studies of Byrne et al.²² The zinc acetate
157 solution was allowed to remain on the substrate surface for 20 s, thus allowing the solution
158 to spread over the entire substrate. After this step, the material was rinsed with a copious
159 amount of ethanol and dried with nitrogen. The zinc acetate coating procedure was
160 performed five times consecutively in each sample, and thereafter the substrate containing
161 zinc acetate was annealed in atmospheric air at 350 °C for 30 min.

162 The next step was to prepare the chemical bath. For this procedure,¹⁹ 100 mL of a
163 0.02 mol L⁻¹ zinc nitrate solution was slowly added to 100 mL of a 0.8 mol L⁻¹ NaOH
164 solution with vigorous stirring. This mixture was heated to 70 °C and ZnO seeded
165 substrates were submerged into the solution and maintained at this controlled temperature
166 with stirring for 25 minutes. Subsequently, the substrates were removed from the bath,
167 washed with deionized water and dried with a nitrogen stream.

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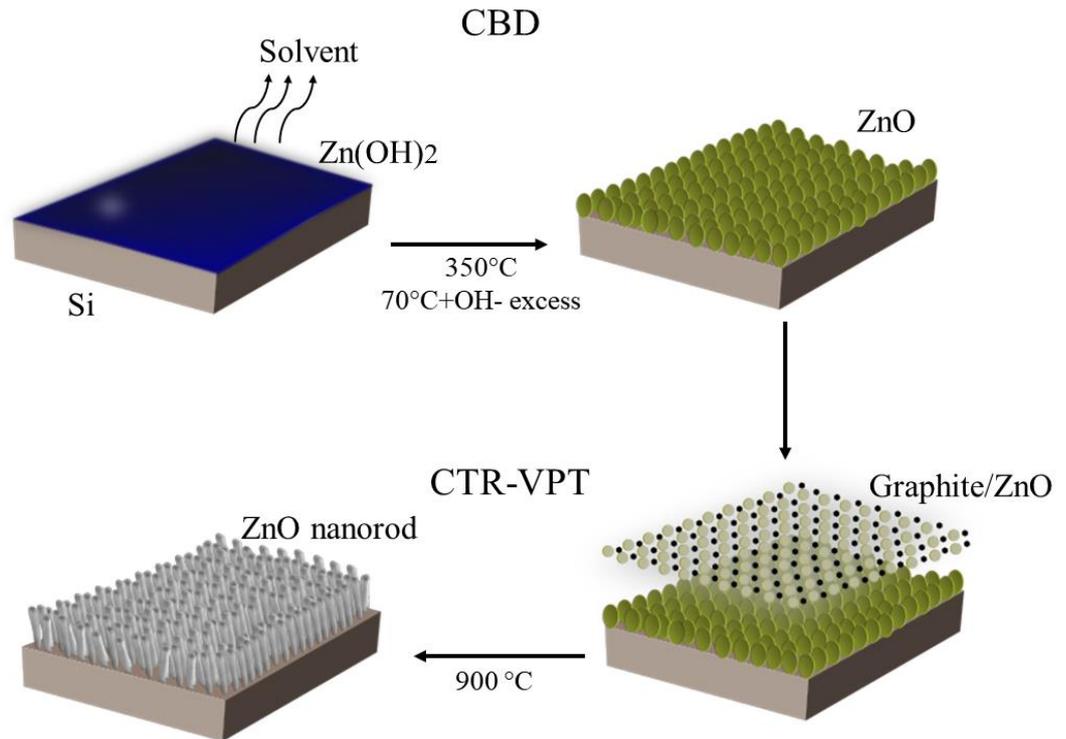
170 2.2.2. Carbothermal reduction vapour phase transport deposition (CTR-VPT)

171 In this procedure, 0.06 g of ZnO powder was mixed with an equal mass of graphite
172 until they formed a homogeneous gray powder and this was then distributed evenly into
173 an alumina boat. The substrates containing ZnO nanorods deposited by the CBD
174 technique described above (which are now used as a seed layer for CTR-VPT growth)
175 were placed directly over the graphite/ZnO mixture (in identical symmetrical positions)
176 so that when it was placed in the furnace, the vapors from the mixture rise directly onto
177 the CBD ZnO nanorod covered Si surface. The boat containing the substrate was then
178 placed inside a quartz tube and positioned at the centre of a horizontal single zone tube
179 furnace and was heated at 900 °C under a 100 sccm argon flow and maintained at this
180 temperature for 1 h. After this period, the material was cooled to room temperature and
181 removed from the furnace. A schematic diagram of the ZnO nanorod synthesis procedure
182 is given in Figure 1.

183

184

185



186

187 **Figure 1:** Schematic representation of growth mechanisms of ZnO nanorods in the CBD
 188 and CTR-VPT processes, showing the formation of the p-Si/n-ZnO
 189 heterojunction.

190

191 2.3.Characterization

192 The morphologies of the deposits were examined using SEM (Karl-Zeiss EVO
 193 series). The crystal phases of the synthesized samples were determined by XRD analysis
 194 using a Bruker AXS D8 Advance Texture Diffractometer with CuK α radiation ($\lambda =$
 195 1.541874 Å) over the 2θ range 10–80°. Material surface composition was analyzed by
 196 XPS using a VG Microtech electron spectrometer at base pressure of 1×10^{-9} mbar using
 197 a conventional Al K α ($h\nu = 1486.7$ eV) x-ray source. The pass energy of the analyser was
 198 set at 20 eV, yielding an overall resolution of 1.2 eV. All peak analysis presented in this
 199 study was performed using AAnalyzer curve fitting software program version 1.20. The
 200 calibration of the binding energy scale was performed with the C1s line (285 eV) from
 201 the adventitious carbon contamination layer.

202 Contact angle (CA) measurements were performed using an FTA-200 contact
203 angle analyzer (First Ten Angstroms, USA) by imaging a droplet of water that was
204 dispensed onto the ZnO nanorod deposit surface. The average CA for distilled water was
205 determined in a progression of ten estimates for each electrode. The captured images were
206 then analysed using FTA32 software.

207 The linear scan voltammetry plots for measuring photocurrents in order to analyze
208 the photoactivity of the synthesized ZnO electrodes were carried out using a VersaStat II
209 potentiostat/galvanostat (Princeton Applied Research) controlled via Echem software
210 using a three-electrode configuration with a Pt network counter electrode, a Ag/AgCl/KCl
211 3 mol L⁻¹ reference electrode, and the synthesized ZnO electrodes as working electrodes.
212 The working electrode was 6.5 cm² in active area. The parameters of voltammetry were
213 as follows: potential range = - 0.5 to + 3.5 V; equilibrium time: 15 s; scanning speed: 5
214 m Vs⁻¹. The electrolyte was 0.05 mol L⁻¹ Na₂SO₄ solution. The photoresponse of the ZnO
215 electrodes were measured as UV-ABC radiation source using an 80 W HPL-N, high-
216 pressure mercury vapor lamp (222-578 nm, with maximum emission at 254 nm, Orsan)
217 at a flux of 3.71x10¹⁹ photons s⁻¹ experimentally determined by chemical actinometry.³⁷
218 The system was kept under constant stirring and the lamp was immersed in the 350 mL
219 electrolyte solution inside a quartz tube with water inlet and outlet, allowing cooling of
220 the system (keeping the working solution at ~ 25°C) with the aid of a thermostatic bath.

221

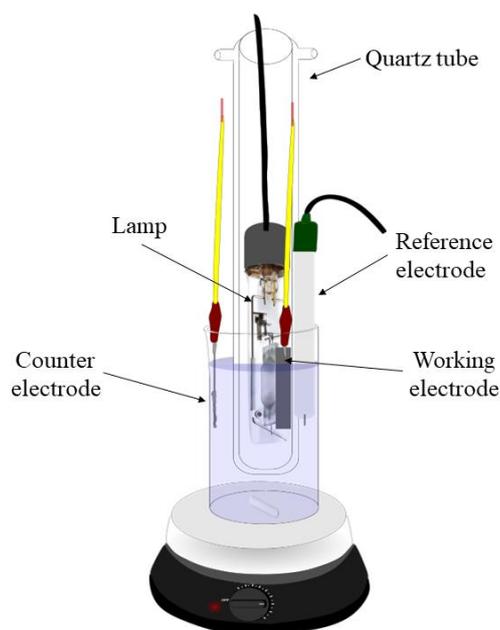
222 *2.4. PEC degradation*

223 The PEC activity of the ZnO photoelectrodes was evaluated by degrading MB
224 solution under UV-ABC irradiation (the same lamp used to investigate the photoresponse
225 of the ZnO electrodes, see details above). The photoelectrodegradation experiments were
226 carried out in a PEC reactor, composed of three electrodes: a Pt network as counter

227 electrode, ZnO nanorods synthesized by the CBD technique and/or ZnO nanorods grown
228 by the CTR-VPT technique as working electrodes and a Ag/AgCl/KCl 3 mol L⁻¹, as the
229 reference electrode. The reactor was filled with an aqueous solution of 20 mg L⁻¹ of MB
230 prepared in 350 mL of Na₂SO₄ (0.05 mol L⁻¹). A positive bias potential of 0.6 V³² was
231 applied by a potentiostat (VersaStat II), for a period of up to 180 min. The schematic of
232 the PEC reactor used is illustrated in Figure 2.

233 For comparison, a PEC experiment was also conducted using samples of pure p-
234 type Si (the support material) as working electrodes, as well as an experiment using direct
235 photolysis and experiments using only PC, without application of an external potential.
236 All control experiments were performed under the same conditions described above.

237 The the conversion rate was examined by measuring the absorbance of the MB at
238 664 nm on a Unicam UV–Vis spectrophotometer. To monitor MB concentration, a
239 calibration curve was obtained in the 0.25-25 mg L⁻¹ range, with Abs (a.u.) = 0.0027 +
240 0.12813 [MB, mg L⁻¹]; R = 0.999, SD = 0.024.



241

242 **Figure 2:** Scheme of the PEC reactor composed of three electrodes.

243

244 3. Results and Discussion

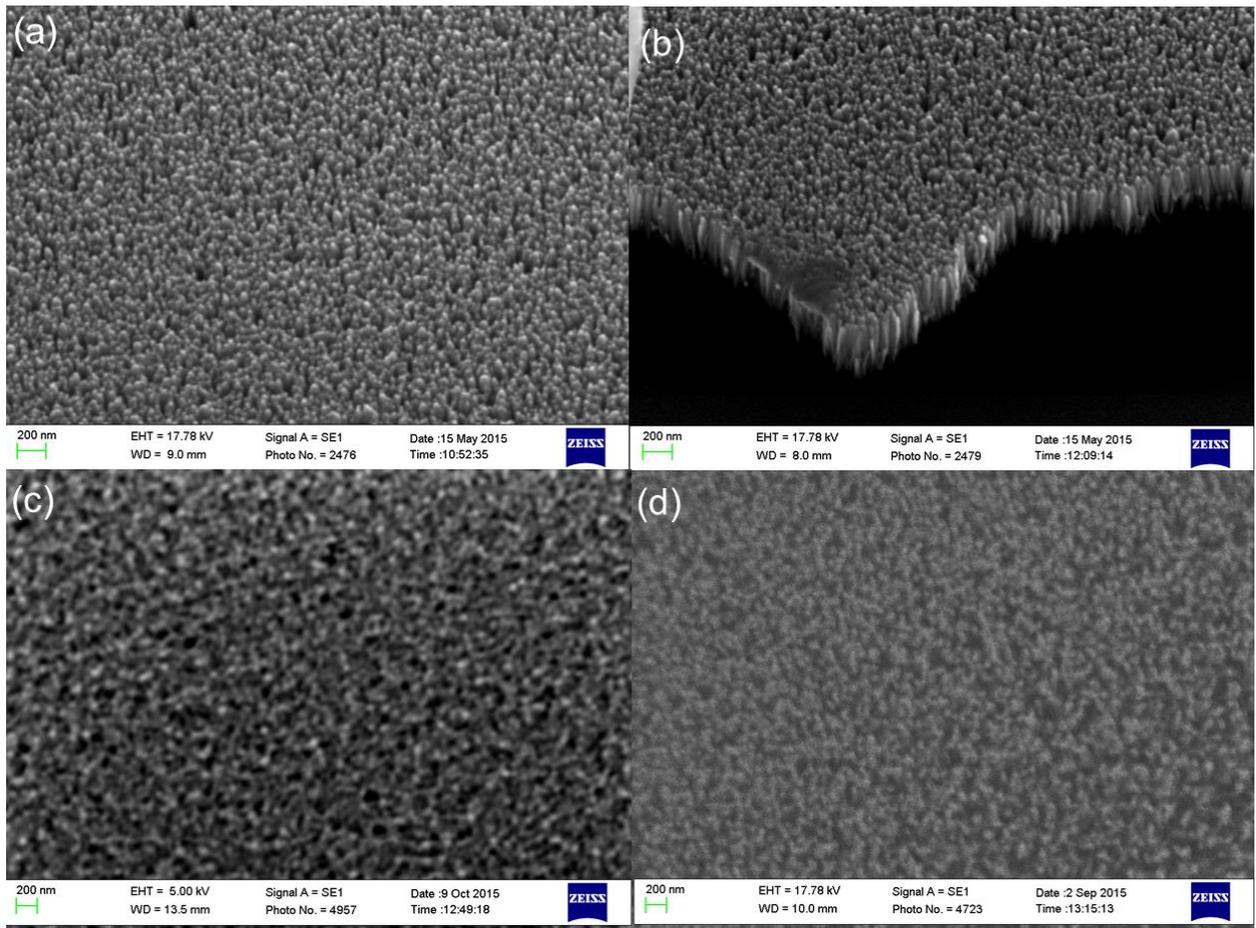
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246 3.1. Characterization results

247 Samples obtained using both methods of ZnO synthesis gave rise to nanorods on
248 the Si substrates. Figure 3 shows SEM images of ZnO nanorod films deposited by CBD
249 for three different samples. Figures 3(a) and (b) show plan and cross-sectional views for
250 one sample (labelled C1). From these images we clearly observe well-aligned ZnO
251 nanorod arrays with uniform diameters and uniform coverage over the Si substrate. The
252 morphology of samples C2 and C3 are shown in Figures 3(c) and (d) and reveal that the
253 nanorods are less obviously distinct and more densely packed.

254 Slight variations in morphology between samples of the sort seen in Figure 3 were
255 also observed by Maryam et al.³⁸ in a study of synthesis of ZnO nanorods on glass
256 substrates. The differences are quite minor and are likely due to small variations in the
257 deposition of the initial seed layer of ZnO deposited by drop coating, a step that proved
258 to be very important for obtaining well-defined ZnO nanorods by the CBD technique.

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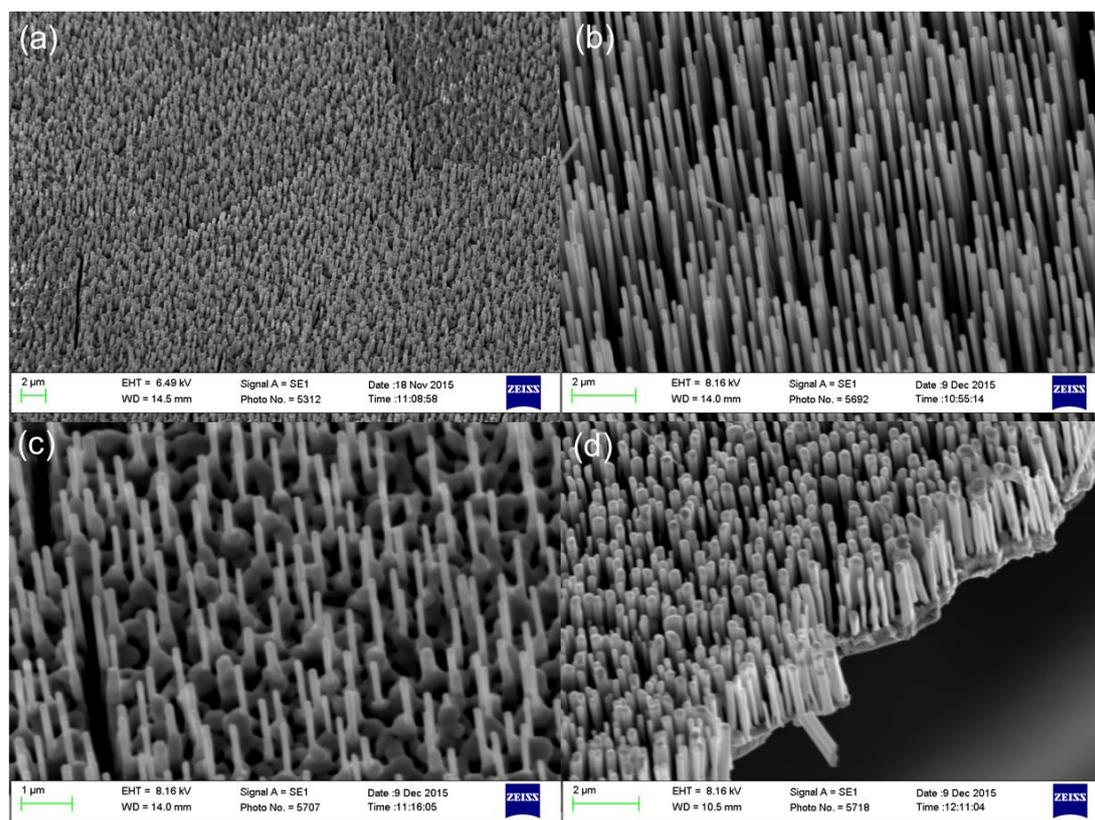
261 **Figure 3:** SEM images of the ZnO nanorods grown by the CBD method showing (a) a
 262 plan view and (b) a cross-sectional view of sample C1 and plan views for
 263 samples C2 (c) and C3 (d).

264

265 Following CBD deposition, CTR-VPT growth was performed on some samples.
 266 Figure 4 shows plan and cross-sectional view SEM images of ZnO nanorods films after
 267 CTR-VPT growth. The CTR-VPT step produced longer narrow, well-aligned ZnO
 268 nanorods with more uniformly diameters a top the original CBD seed layer.

269 As can be seen in Figure 4, the ZnO nanorods are uniformly distributed on the Si
 270 substrates. The average diameter of ZnO/CTR-VPT nanorods at the center of the sample
 271 is 158 ± 25 nm and their average height is 4.7 ± 0.4 μm . At the edges of the ZnO arrays
 272 the average diameter is 227 ± 68 nm and the average height is 1.6 ± 0.2 μm .

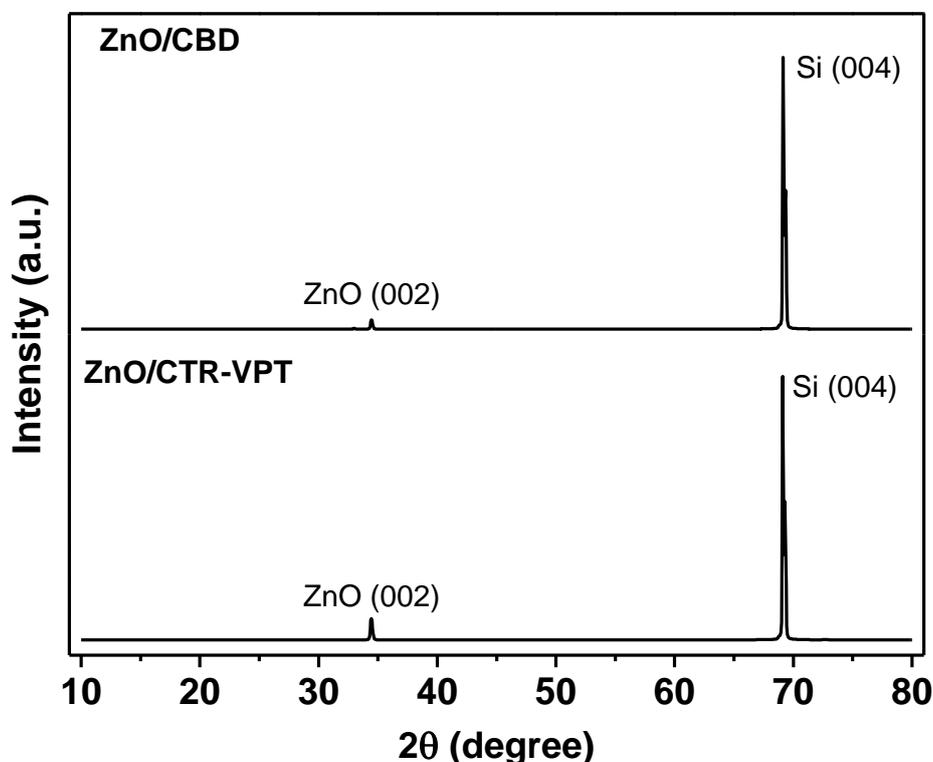
273 The rods are narrower and longer in the center of the sample and, as they approach
274 the edges, they become wider and shorter. These variations of the size of the ZnO
275 nanorods likely result from a variation in supersaturation of the Zn^{2+} growth species
276 during the synthesis process, as has been reported previously.³⁹
277



278
279 **Figure 4:** SEM images of the ZnO nanorods grown by CTR-VPT. (a) Plan view and (b-
280 d) 30° view.

281
282 Figure 5 shows XRD results for the deposited ZnO nanorods grown by both
283 methods. The predominant diffraction peaks observed at a 2θ value of 34.45° correspond
284 to reflections from (002) planes of wurtzite hexagonal ZnO.⁴⁰⁻⁴² No diffraction peaks
285 corresponding to metallic Zn were found in any region of the electrode. This sharp and
286 dominant (002) diffraction peak indicates the nanorods are well crystallized and highly
287 oriented with their c-axes orientation normal to the substrate surface.^{43,44} The intense peak

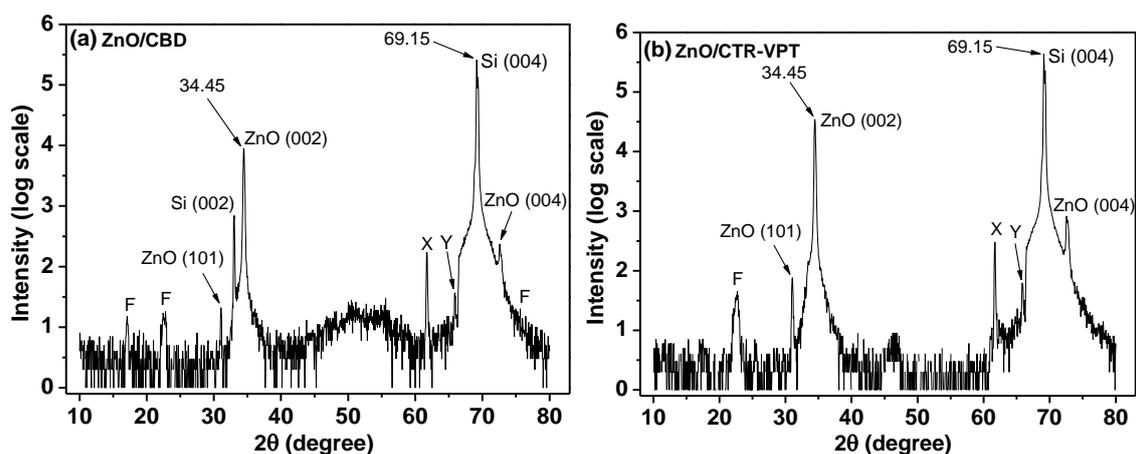
288 located at 69.15° corresponds to the Si peak from the (004) Si planes used as the
 289 substrate.^{45,46}
 290



291
 292 **Figure 5:** XRD patterns of ZnO nanorod arrays grown on silicon, displaying two
 293 dominant peaks, associated with the ZnO (002) and Si (004) reflections, at 34.41° and
 294 69.15° , respectively.

295
 296 In order to observe the low intensity peaks in the diffractogram, the spectra was
 297 plotted on a logarithmic y-scale.⁴⁷ Figures 6(a) and (b) shows the XRD diffractograms of
 298 the ZnO/CBD and ZnO/CTR-VPT samples on a logarithmic y-scale. It can be observed
 299 that in addition to the two most intense peaks at 34.45° and 69.15° , there are reflections
 300 from the (004) and (100) planes of ZnO⁴⁵ for the samples synthesized by both methods,
 301 as well as the nominally forbidden Si(002) reflection for the sample synthesized by CBD
 302 method, due to double diffraction, whose intensity depends on the azimuthal angle, ϕ ,

303 and which hence is not seen with equal intensity in all samples.⁴⁸ Some small peaks,
 304 labelled as F, are also observed and are attributed to the adhesive tape used to mount the
 305 samples at the time of analysis. Gray et al.⁴⁵ also report the presence of the same F peaks.
 306 According to the studies of Gray et al.⁴⁵ and Kumar et al.⁴⁹ the point marked as X is due
 307 to $K\beta$ radiation at $\sim 61.7^\circ$ from the X-ray tube and the feature marked as Y is due to
 308 tungsten $L\alpha$ radiation at $\sim 65.7^\circ$ from contamination of the x-ray tube Cu target by the
 309 electron gun filament. No other peak-related impurities were observed in the pattern
 310 which confirms the presence of pure wurtzite phase ZnO nanostructures.
 311



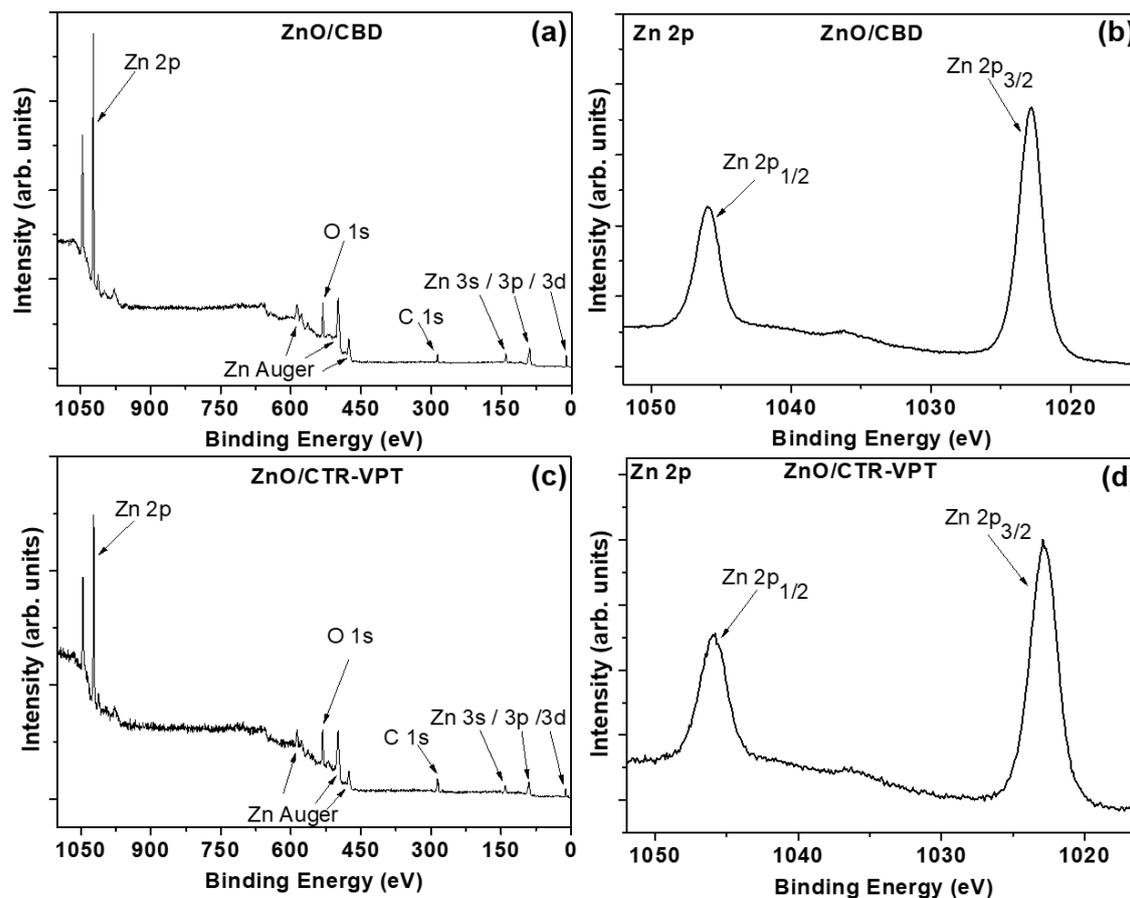
312
 313 **Figure 6:** XRD patterns of ZnO samples on a logarithmic y-scale, to enhance the
 314 visibility of low intensity peaks.
 315

316 Figure 7 shows the results of XPS measurements of ZnO nanorods. Figures 7(a)
 317 and (c) show XPS survey spectra of the samples grown by CBD and CTR-VPT,
 318 respectively, confirming the presence of elements Zn, O and C. No other elements are
 319 seen. The presence of C is due to atmospheric contamination (seen on all samples exposed
 320 to ambient conditions). Thus, the results confirm the high purity of the ZnO nanorods and
 321 is in accordance with the XRD analysis.

322 Figures 7(b) and (d) shows the Zn 2p core level emissions from ZnO nanorods
323 grown by CBD and CTR-VPT, respectively. For the ZnO/CBD sample, the doublets
324 corresponding to the $2p_{3/2}$ and $2p_{1/2}$ photoelectron core level peaks are centrally
325 positioned at 1022.8 eV and 1046 eV, respectively, in the spectrum. In case of ZnO/CTR-
326 VPT the electronic states of Zn $2p_{3/2}$ and Zn $2p_{1/2}$ were observed at the binding energies
327 of 1022.9 eV and 1045.9 eV, respectively. These values agree within the energy
328 resolution of the system. For the ZnO/CBD sample the binding energy distance between
329 the two spin orbit split peaks is 23.2 eV while in case of ZnO/CTR-VPT sample the spin
330 orbit separation is 23 eV, again in agreement within the system's energy resolution, and
331 demonstrating that the Zn species exist in the Zn^{2+} chemical state, consistent with
332 previous work in the literature and with reference data for ZnO.⁵⁰⁻⁵³

333

334



335

336 **Figure 7:** XPS spectra for the ZnO/CBD (a, b) and ZnO/CTR-VPT samples (c, d). (a, c)

337

XPS survey spectra, (b, d) high resolution Zn 2p XPS spectrum.

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Figure 8 shows the XPS spectra in the O 1s region for the ZnO/CBD sample. The O 1s signal can be deconvoluted into two Gaussian peaks, at binding energies of 531.6 eV and 533.04 eV, commonly called O(1) and O(2), respectively. The peak at the lower binding energy is assigned to O²⁻ ions in the O-Zn bonding matrix of the hexagonal ZnO wurtzite structure.^{50,52-56} The other peak at the binding energy of 533.04 eV is associated with O²⁻ that is present in the oxygen deficient regions^{53,57} or OH species on the surface of ZnO nanorods.^{52,55,58}

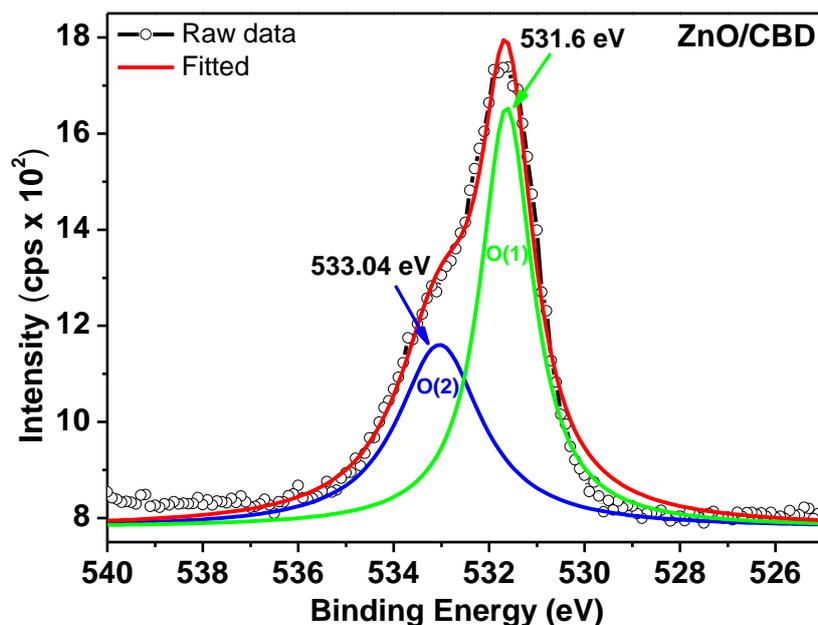


Figure 8: Oxygen 1s XPS spectra of ZnO nanorods grown by CBD.

To investigate the wettability of the ZnO nanorods, a water CA measurement was carried out.^{59,60} The ZnO nanorod arrays grown by CBD showed the highest water contact angle of about $92^\circ \pm 4^\circ$. In contrast, ZnO/CTR-VPT samples show water contact angle of $64^\circ \pm 8^\circ$, indicating an increase in the hydrophilicity.⁶¹ The wettability is an important property of solid surfaces that depends on the chemical composition, energetics and geometric surface structures. In this case, ZnO/CTR-VPT samples have a lower contact angle, possibly due to their well-organized, and quite long, nanorod morphology, leading to greater hydrophilicity.^{26,60}

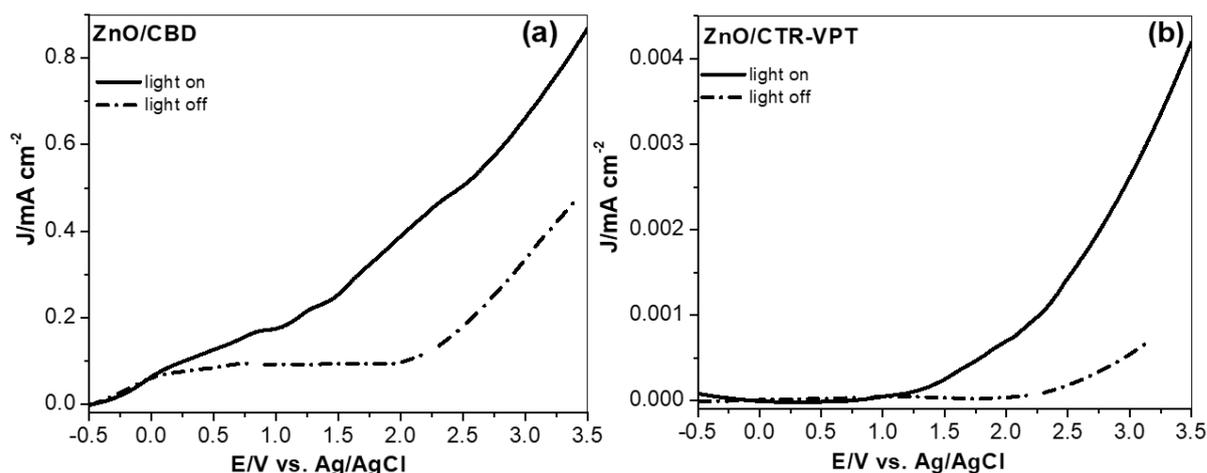
Figures 9(a) and (b) show the photocurrent densities of ZnO/CBD and ZnO/CTR-VPT photoanodes operating under dark and illuminated conditions. The results indicate that ZnO nanorod arrays formed by both techniques respond to UV-ABC irradiation. The current response of both films in the dark is very weak. However, when the experiment is performed under illumination, the current increases strongly, showing that synthesized ZnO nanorods are a good photocatalyst for PEC.⁵ It can be seen that UV-ABC irradiation

364 obviously increases the current when the potential is higher than 0 V. This is attributed to
365 photogenerated electrons on the ZnO nanorods driven to the counter electrode by the
366 application of a positive potential. Thus, the recombination of the photogenerated
367 electron–hole pairs is hindered and increased photocurrent is generated.^{5,18}

368 The photocurrent density of the ZnO nanorods prepared by CBD was observed to
369 be highest ($\sim 0.87 \text{ mA cm}^{-2}$ at +3.5 V); it was about two hundred times that of the
370 ZnO/CTR-VPT sample at +3.5 V. This result indicates that the ZnO nanorod arrays
371 prepared by CBD possess excellent PEC response under UV illumination.

372 Due to the fact that ZnO grown by CTR-VPT presented a morphology with more
373 organized nanorods, it was initially expected that these samples would present better
374 charge transport, leading to a higher photocurrent density. However, the photocurrent
375 across the ZnO/CBD sample is much higher. The enhancement of the photocurrent of the
376 ZnO/CBD photocatalyst compared to ZnO/CTR-VPT samples may have a number of
377 origins. Firstly the higher growth temperatures used for CTR-VPT growth can lead to an
378 increased thickness SiO_2 layer between the Si and ZnO, retarding current flow.⁶² A further
379 contribution may come from heterojunction formation between the p-Si (substrate) and
380 the CBD ZnO layer with lower thickness; thus, when the p-Si and the n-ZnO make
381 electrical contact with each other their Fermi levels will equalise, leading to the shift of
382 their conduction and valence bands;³⁵ Another possible source for the lower photocurrent
383 of the ZnO/CTR-VPT sample is the presence of defects on surface of the ZnO/CTR-VPT,
384 which may lead to relatively low efficiency separation of the photogenerated carriers.⁶⁰
385 XPS data presented previously indicate slight differences in surface chemistry between
386 the CBD and CTR-VPT samples.

387



388

389 **Figure 9:** Current density-voltage curves of the ZnO nanorod samples in the dark and
 390 under UV-ABC irradiation recorded in a $0.05 \text{ mol L}^{-1} \text{ Na}_2\text{SO}_4$ electrolyte at a
 391 scan rate of 50 mV s^{-1} .

392

393 3.2. Investigation of PEC efficiency of ZnO nanorods

394 The PEC degradation of MB solution under UV-ABC light irradiation was used
 395 to evaluate the PEC activities of ZnO/CBD and ZnO/CTR-VPT electrodes under the
 396 experimental conditions indicated previously. The results of degradation experiments are
 397 shown in Figure 10(a). About 43% of MB was degraded in 180 min when the ZnO/CTR-
 398 VPT electrode was used for the PEC process, while it improved to 53% when using the
 399 ZnO/CBD photoelectrode. By comparison, the removal rate using the PC technique is
 400 only 41% and 39% using ZnO/CBD and ZnO/CTR-VPT, respectively.

401 As mentioned in the introduction, Han et al.³⁰ prepared ZnO nanorod arrays on
 402 Ag by CBD and evaluated the PEC performance of the electrode for the removal of
 403 rhodamine B (RhB). Applying a potential of 0.4 V and using 0.1 mol L^{-1} of Na_2SO_4 as
 404 support electrolyte, the authors obtained 38% RhB removal after 5 h of treatment. This
 405 study demonstrates that our results are comparable with those found in the literature for
 406 similar morphology ZnO nanomaterials.

407 Figure 10(b) shows the variation in the absorption spectra from 200 to 800 nm of
408 MB collected at different time intervals during the PEC degradation experiment using a
409 ZnO/CBD sample as a catalyst. The absorption spectrum of MB shows two peaks of lower
410 intensities at 246 and 292 nm and a more intense peak at 664 nm. The peaks at 246 and
411 292 nm are due to the substituted benzene ring structures while the peak at 664 nm is
412 attributed to the auxochrome group of MB.⁶³ It can be seen that the absorbance peaks
413 decreased gradually as the exposure time increases from 0 to 180 min. The visual
414 appearance of the MB solution changed from dark blue to nearly colourless over this
415 period.

416 A control experiment was conducted by studying the degradation of this dye under
417 UV-ABC irradiation in the absence of electrodes. In this case, about 39% removal of MB
418 was obtained. As a comparison, the PEC process using bare Si as the working electrode
419 was also carried out in order to verify the influence of the substrate in the degradation of
420 the MB. This blank experiment showed that the degradation rate is 36% in 180 min. It is
421 observed that the degradation rate was lower than that found in direct photolysis,
422 indicating that the PEC activity of Si can be neglected. One hypothesis to elucidate the
423 lower degradation rate in the presence of Si compared to direct photolysis would be that
424 MB molecules adsorb on the surface of the material and remain adsorbed due to the low
425 conductivity of the pure Si and the insulating native oxide (SiO₂).

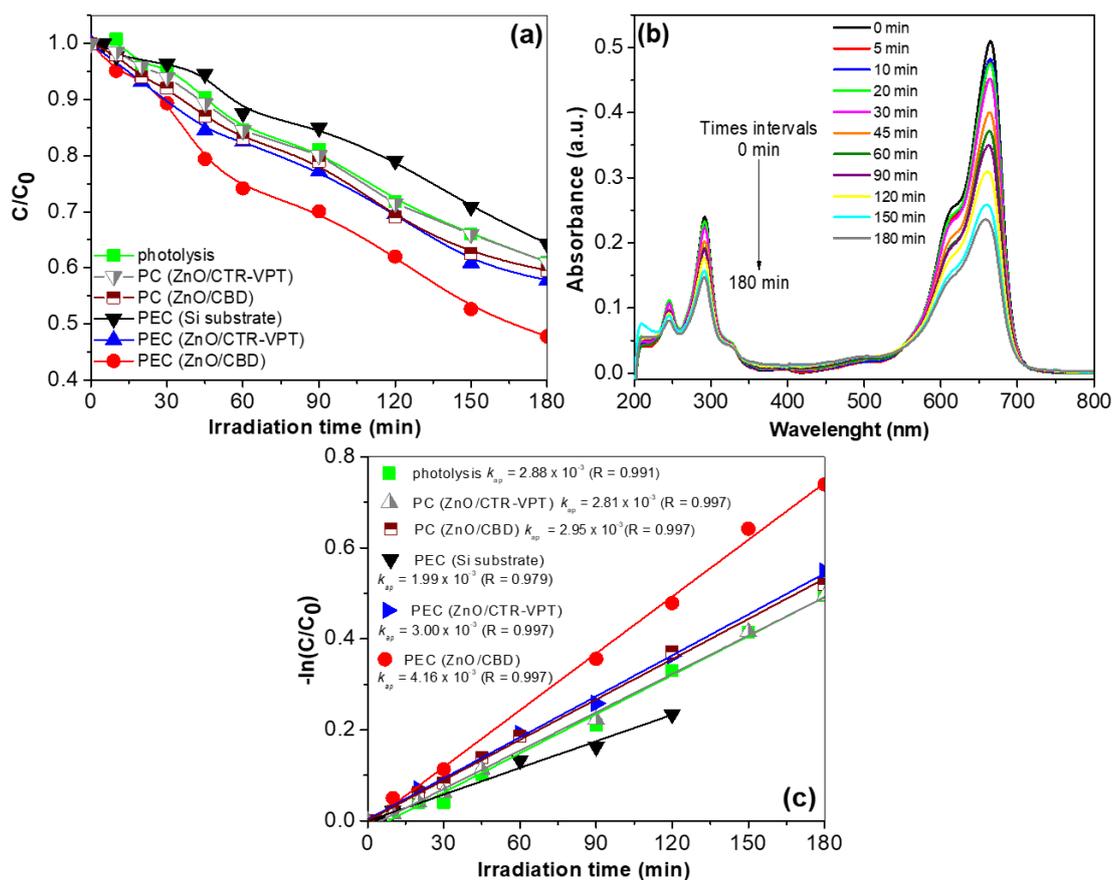
426 The degradation rates of MB were analysed using the Langmuir–Hinshelwood
427 pseudo-first order kinetics model. Following this model, the degradation rate constants
428 (k_{ap}) are calculated from the linear fit extracted from the $-\ln(C/C_0)$ vs. Time data (Figure
429 9(c)). As shown in Figure 10(c), the magnitude of k_{ap} shows the following ordering;
430 ZnO/CBD > ZnO/CTR-VPT = direct photolysis > pure Si. It is apparent that the kinetic
431 constant of PEC using a ZnO/CBD photoelectrode is the highest, and is 1.4 times more

432 effective than PC.

433 Comparing the processes of direct photolysis and PEC degradation using a
434 ZnO/CTR-VPT electrode as catalyst, it can be seen that there was no significant
435 difference in the degradation rate of the MB, which shows that the synthesized ZnO does
436 not demonstrate PEC activity under the applied conditions. The low photoactivity of this
437 electrode in comparison to the electrode grown by CBD is due to the low current density
438 seen in the potential photocurrent curves (Figure 10(b)), $\sim 4.18 \mu\text{A cm}^{-2}$ at 3.5 V. SEM
439 images of the ZnO/CTR-VPT surface before (Figure 4) and after (Figure 11) the PEC
440 process show that the applied potential may also have caused a physical degradation of
441 the surface of the oxide, leading to a further reduction of its photocatalytic activity.

442 Other factors may also have influenced the PC and PEC responses, leading to
443 improved results for ZnO/CBD electrodes compared to ZnO/CTR-VPT electrodes. The
444 results of CA measurements, show that the ZnO/CBD electrode presented a higher
445 contact angle, consequently it presents a greater wettability, which may lead to a better
446 interaction between the surface and working solution and, consequently higher
447 production of hydroxyl radicals (HO^\bullet), increasing the photocatalytic response. SEM
448 images show that the CBD-grown nanorods are smaller and consequently the surface area
449 is larger; it is well known that a photoconductor with a relatively larger surface area
450 in general tends to show better catalytic response.

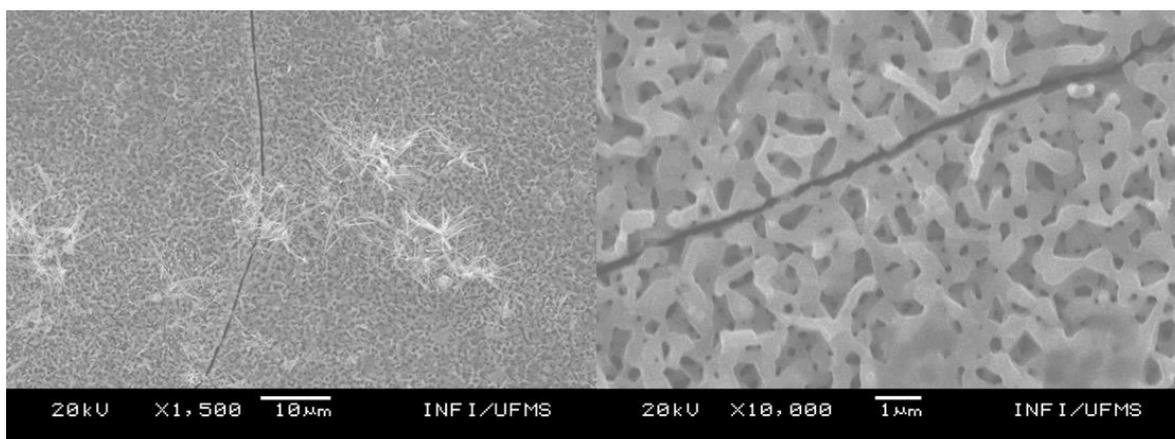
451



452

453 **Figure 10:** (a) MB degradation curves under UV-ABC irradiation; (b) Absorbance
 454 spectra of MB with respect to time subjected to PEC treatment with ZnO/CBD
 455 electrode at 0.6 V; (c) Corresponding kinetics fitting curves.

456



457

458 **Figure 11:** SEM images of ZnO nanorods grown by CTR-VPT after the PEC process.

459

460 As mentioned in the introduction, there are few reports of the application of ZnO
461 nanorods for PEC degradation of contaminants compared to TiO₂. We now discuss the
462 reports mentioned in the introduction in more detail, in order to allow direct comparison
463 with our results. Fan et al.⁵ evaluated the efficiency of ZnO films deposited on titanium
464 plates by liquid phase deposition (LPD) as photoanodes for the removal of p-nitrophenol
465 by PEC. The authors obtained 91% of p-nitrophenol removal after 180 min of irradiation
466 in optimized conditions. Hunge et al.⁶ synthesized ZnO thin films on glass and fluorine
467 doped tin oxide (FTO) coated glass substrates by spray pyrolysis and evaluated the
468 efficiency of these electrodes for PEC degradation of terephthalic acid. The degradation
469 percentage of terephthalic acid using ZnO photoelectrode reached up to 91% under
470 ultraviolet illumination after 400 min. Liu et al.³¹ synthesized CdS-Coated ZnO nanorods
471 arrays by a two-step method. Firstly, ZnO nanorod arrays were grown under hydrothermal
472 conditions on an ITO substrate and secondly, a coating of CdS on the surface of the ZnO
473 was realized by a successive ionic layer adsorption and reaction (SILAR) method and the
474 authors then evaluated the efficiency of the photoanodes for PEC degradation of phenol.
475 In the PEC process, about 80% degradation of 100 mg L⁻¹ phenol solution is achieved
476 within 150 min under visible light irradiation. Lin et al.³² evaluated the PEC degradation
477 of paracetamol using ZnO nanorod-array electrodes on FTO glass via a hydrothermal
478 method. The authors obtained 62% paracetamol removal after 20 hours. Suryavanshi et
479 al.³³ prepared ZnO thin films on glass and FTO coated glass substrates by spray pyrolysis
480 and evaluated the efficiency of these electrodes for PEC degradation of of benzoic acid
481 (BA) and methyl blue (MB) dye under UV radiation. The results of this study showed
482 65.7% degradation of BA and 98.1% of MB within 400 min and 120 min, respectively.
483 Sarwar et al.³⁴ investigated the removal of textile dyeing effluents with voltage-assisted
484 PC activity using carbon fabrics containing ZnO nanorods as photoanodes. In this study,

485 ZnO nanorods were grown by a seeding technique followed by a hydrothermal process.
486 The study concludes that carbon fabric treated with ZnO nanorods can be successfully
487 utilized for enhanced decolorization of dye contaminated wastewater, providing an
488 environmental friendly solution for the treatment of effluents generated by textile, leather
489 and other industries. Our results are consistent with these literature reports and allow us
490 to claim that ZnO nanorods deposited on Si substrates by CBD show good potential for
491 use as a working electrode in PEC, and that the use of Si substrates, in addition to the
492 facile CBD synthesis process provides some advantages compared to the other methods
493 described in terms of scalability and cost.

494

495 **4. Conclusions**

496

497 In this study, we successfully fabricated well-aligned ZnO nanorods on Si with a
498 geometrical area of $\sim 6.5 \text{ cm}^2$ by CBD and CTR-VPT.

499 The ZnO nanorods synthesised have a hexagonal wurtzite structure and are
500 textured normal to the substrate surface, as shown by XRD and SEM data. The
501 nanostructures synthesized by CTR-VPT resulted in a more organized structure with
502 narrower and longer nanorods.

503 XPS data indicated the presence of pure ZnO deposited on the Si substrate, while
504 contact angle measurements revealed that nanorods synthesized by CTR-VPT have
505 greater wettability compared to ZnO synthesized by CBD. The photoactivity of
506 ZnO/CBD is larger than that of ZnO/CTR-VPT. After 180 min illumination, the PEC
507 degradation of MB concentrations using ZnO/CBD reached 53%, about 1.2 times that of
508 ZnO/CTR-VPT samples. The degradation process follows pseudo-first order kinetics and
509 k_{ap} values have been extracted.

510 This work demonstrates that more ordered ZnO nanorods were not the best
511 morphology for applications in PEC, and we show that the CTR-VPT process does not
512 produce ZnO nanorods optimised for applications in PEC. The simpler and less expensive
513 CBD technique by itself produces samples which show better performance for PEC
514 applications and which are also more chemically robust.

515

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517

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527

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